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A STUDY OF ELECTRICAL PROPERTIES OF DISLOCATION ENGINEERED SI PROCESSED BY ULTRASOUND

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**Abstract** 

This work presents experimental study of electrical properties of dislocation

engineered Si p-n junction before and after influence of ultrasound waves. We have studied

current-voltage characteristics in the dark and upon illumination for forward and reverse

biases before and after ultrasound processing. By fitting the theoretically established current-

voltage dependence to the experimentally measured ones the diode ideality factor and

saturation current have been estimated. It is found that current transport through the

dislocation-engineered Si p-n junction can be controlled by generation-recombination or

tunneling recombination mechanisms. Ultrasound is found to modulate electrical properties of

the dislocation engineered Si.

**Keywords:** dislocation engineering, current transport, ultrasound treatment

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### 1. Introduction

Since the discovery [1-3] of luminescent feature, study of dislocation engineered (DE) Si is one of the interesting scientific topics. Distinct from bulk Si, luminescence intensity of DE Si is found to increase with increasing the temperature from low to T=300 K. Here the strain field around dislocations plays an important role. On the one hand it modulates band structure, provides carrier confinement, and suppresses diffusion of charge carriers to the surface, thus enhancing radiative recombination and luminescence intensity of the DE Si. On the other hand, the strain field can be the source for aggregation of defects and impurities, which can enhance non-radiative recombination of charge carriers and thus partially suppress radiative recombination and luminescence. To solve the problem, passivation by hydrogen [4] and annealing at high temperatures >200 °C can be used. However, the high temperature annealing can result in quenching the luminescence of DE Si [5]. Thus search of the alternative ways of annealing at moderate temperatures without destroying luminescence properties of DE Si is an important prolem. Ultrasound treatment (UST) could be the alternative way [6-8]. According to scientific literature, UST has successfully been used so far for metal cluster engineering in ion implanted silicon dioxide [9], defect engineering in Si p-n junctions [10], silicon based impact ionization avalanche transit-time (IMPATT) diodes [11, 12], power Schottky diodes and Zener diodes [13, 14], extension of spectral sensitivity of the AlGaAs/GaAs solar cells toward short-wave lengths [15], modification of radiation-induced defects in Si [16], etc. The aim of this paper is to study effect of ultrasound on electrical current transport in dislocation engineered Si p-n junction.

# 2. Experimental details

The samples studied in the paper are fabricated by boron implantation into Czochralski (CZ) P-doped n-type Si (2-7  $\Omega \times$ cm) with contact area 1.76 mm<sup>2</sup> and width 1.00 mm. The boron-

implantation induced dislocation loops located at ~300-700 nm from the Si surface have been detected by TEM analysis [17-19]. Total number of loops is ~3-6×10<sup>9</sup> cm<sup>2</sup>. More details on preparation of samples can be found somewhere else (see, e.g., Refs. [17-20]). Three groups of samples denoted by A1-A4, B1-B4, and C1-C4 have been studied (Table I). S impurities have been implanted into the A1-A4 samples of dose 10<sup>14</sup> cm<sup>-2</sup> with annealing temperature of 1000 °C and B1-B4 samples of dose 10<sup>13</sup> cm<sup>-2</sup> with rapid thermal annealing at 1100 °C. The C1-C4 samples do not contain S impurities.

Sulphur related luminescence of Si has been studied before (see, e.g., Refs. [21-23]), however, no room temperature luminescence has been found. After the dislocation engineered Si light emitting diode was first fabricated [3] effect of additional doping with sulphur impurities was also studied [20]. Below we study effect of UST on electrical properties of the sulphur doped dislocation engineered Si. UST of the samples has been performed from the contact side by ceramic piezoconvertor. Frequency of the UST was 2.4 MHz. The UST has been done in alcohol. Samples have been treated by ultrasound of powers 0.5 and 1.0 W/cm<sup>2</sup> within 15 and 30 min (Table I).

Dependence of electrical current density J on applied voltage V has been studied for forward and reverse biases. The results have been fitted with formula

$$J = J_0 \left[ \exp \left( \frac{qV}{\beta kT} \right) - 1 \right] . \tag{1}$$

From the fitting the diode ideality factor  $\beta$  and saturation current

$$J_0 = q \frac{n_i^2}{N_d} \sqrt{\frac{D_p}{\tau_p}} \,. \tag{2}$$

has been determined. Here T is the temperature, k is the Boltzmann constant, q is the electron charge,  $n_i$  is the intrinsic carrier concentration,  $N_d$  is the concentration of shallow donors,  $D_p$  and

 $au_p$  are the hole diffusion coefficient and lifetime, respectively. The measurements have been performed at room temperature.

### 3. Results and discussions

For efficient transfer of energy of the ultrasound to samples the treatment commonly is done in technical alcohol or distilled water, which are not reactive well for the samples considered. Ultrasound treatment of the reference samples in both of the media showed similar result, which indicates that modulation of properties of the samples can be ascribed to ultrasound rather than surface treatment by alcohol. Based on this result UST of the samples has been performed in alcohol.

Since the samples are of DE Si, one can expect that current transport is controlled by carrier recombination. To check validity of the statement and to establish current transport mechanism, current-voltage dependence has been studied in the dark and upon illumination for forward and reverse biases before and after UST. Diodic characteristic has been observed for all the samples considered. The results for the sample B1 have been presented in Fig. 1 (a) and (b) for forward and reverse biases, respectively. The Fig. 1 (a) has been plotted in semilogarithmic scale. It is seen that in the voltage range  $\sim$ 0.1 to 0.4 the J-V curve is almost linear, which means that the J-V dependence is exponential. So, for the voltage range the J-V dependence can be described by Eq. (1). Fitting of the Eq. (1) to the experimental results presented in Fig. 1 (a) has been performed. Analysis shows that the J-V dependencies in forward bias are well described by the Eq. (1). Upon treating by ultrasound the electrical current J increases. It indicates that UST leads to partial inhibition of generation-recombination processes by reducing concentration of recombination active

deep traps decorating the dislocations. This result is consistent with lifetime measurements [20] by open-circuit voltage decay method, which demonstrated increase of carrier lifetime.

By fitting the Eq. (1) to the measured data saturation current  $J_0$  and ideality factor  $\beta$  have been determined (Table I). Analysis of Table I shows that ideality factor is much larger than 2.00 for the samples A1, A2, A3, and B4. It indicates that current transport is basically controlled by tunnelling-recombination mechanism. However, magnitude of  $\beta$  for most of the samples is around 2, which shows that in those samples current transport is controlled by generation-recombination processes. Saturation current  $J_0$  is very large for all the samples considered. The reason can be related to presence of deep level defects located around dislocations.

Upon annealing by ultrasound defect spectra in all of the samples have been considerable modulated. First of all this should leave to modulation of  $J_0$  and  $\beta$ . In samples A3, C3, and C4, magnitude of both  $J_0$  and  $\beta$  has been reduced, which demonstrates UST induced annealing of the deep traps. However, in rest of the samples  $J_0$  and  $\beta$  has been increased, which demonstrates UST induced formation of the recombination active centers.

For convenience of analysis the values of J and V in Fig. 1 (b) have been multiplied to minus one and the figure has been plotted in the first quadrant. Analysis of current-voltage characteristics for reverse bias (Fig. 1 (b)) showed that it can be described by  $6^{th}$  order polynom

$$J(V) \approx J(0) + a_1 \times V + a_2 \times V^2 + a_3 \times V^3 + a_4 \times V^4 + a_5 \times V^5 + a_6 \times V^6,$$
(3)

Here  $a_1$ - $a_6$  are coefficients of the polynomial determined from fitting of Eq. (3) to experimentally measured data. This result shows that the defects responsible for carrier recombination form a continuous energy band in the band gap. One can expect this result because the samples are light-

emitting diodes with dislocation loops. Generation-recombination processes are controlled by dislocations and as is well known [24] the dislocation related bands in Si are continuous.

Analysis of Fig. 1 (b) shows that the J-V dependencies for reverse bias before and after UST differ each from other. The reason can be related to UST-stimulated modulation of recombination activity of the dislocations. As noted in Ref.[25], strain field around dislocations can modulate band structure of Si. So, one the one hand the experimental observation can be the result of the UST-induced transformation of the dislocations, which has lead to changing of their recombination properties. On the other hand, the UST modulation of defects can lead to variation of the concentration and lifetime of the carrier confined in the dislocation loops, which can be released at larger reverse voltages and contribute to the reverse current. So, at present the safest conclusion is that the difference on the reverse J-V dependencies before and UST can be related to UST-induced modulation of recombination activity of dislocations.

Current voltage dependence has been studied not only in the dark, but also upon illumination. From the study it is found that the samples possess photovoltaic (PV) feature. In all the samples photovoltage and photocurrent have been generated. Figure 2 (a) and (b) display the *J-V* curves corresponding to the sample C2 for different illumination intensities (a) before and (b) after UST. As expected, the PV performance of the device structure is too small, because the main aim of incorporation of the dislocation loops is to provide confinement of non-equilibrium charge carriers and to enhance the rate of their radiative recombination rate thus enhancing the intensity of light emission. UST further reduced the PV performance. Similar result has been obtained for the sample C2. For some of the samples the feature has been enhanced.

Such device structures are expected to be sensitive to variations of the illumination intensity.

This feature is available in our samples as well. Figure 3 shows the dependence of short-circuit

current  $J_{sc}$  (Fig. 3 (a)) and open-circuit voltage  $V_{oc}$  (Fig. 3 (b)) on illumination intensity  $\Phi$ . It is seen that at smaller  $\Phi$  both of the parameters characterizing PV feature of the samples increase sharply with increasing  $\Phi$  thus demonstrating high sensitivity of the samples to illumination. At larger values of  $\Phi$  both  $J_{sc}$  and  $V_{oc}$  slightly increase.

As it is well known,  $J_{sc}$  is related to carrier lifetime through the relation [26]

$$J_{sc} = q\Phi \cosh(W/L_p),\tag{4}$$

where  $L_p = \sqrt{D_p \tau_p}$  is the diffusion length of free holes. W is thickness of the base layer, and  $\Phi$  is is the incident photon flux creating free carriers. Based on Eq. (4) one can suggest that the UST-induced enhancement of  $J_{sc}$  (Fig. 3 (a)) can be caused by that of carrier lifetime. This result is consistent with that of Ref. [27] which reported about UST-induced increase of carrier lifetime for the sample A1 (Fig. 3) determined from analysis of the open-circuit voltage  $V_{ac}$  decay transient.

Since the equilibrium and non-equilibrium concentrations of free holes  $p_0$  and p are related to each other through the relation

$$p = p_0 \times \exp\left(\frac{qV_{oc}}{\beta kT}\right) \tag{4}$$

the UST-induced enhancement of  $V_{oc}$  (Fig. 2 (b)) indicates that free carrier concentration has been increased as well. The relation  $p/p_0$  has been estimated using the measured values of  $V_{oc}$  before and after UST. It is found that after UST  $p/p_0$  has been increased ~10<sup>2</sup> times.

### 4. Conclusion

Current-voltage characteristics have been studied for the dislocation engineered Si p-n

junction for forward and reverse biases in the dark and upon illumination before and after ultrasound treatment. Saturation current, ideality factor have been estimated for each of the cases. It is found that the diode ideality factor is in the range 1.8-3.5 before and 2.0-3.8 after ultrasound treatment. This result means that generation-recombination and tunnelling recombination mechanisms of current transport can take place. By analysis of the dark and light current-voltage dependencies before and after ultrasound treatment we show that by ultrasound processing it is possible to modulate the diode ideality factor for the samples except sample C3 by varying the ultrasound power, frequency and duration. Ultrasound can cause transformation of structural defects, which plays important role in transport properties of the dislocation engineered Si.

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#### References

- [1] W. L. Ng, M. P. Temple, P. A. Childs, F. Wellhofer, K. P. Homewood, Appl. Phys. Lett. 75 (1999) 97-99.
- [2] M. A. Lourenco, M. S. A. Siddiqui, R. M. Gwilliam, G. Shao, K. P. Homewood, Physica E 16 (2003) 376-381.
- [3] W. L. Ng, M. A. Lourenco, R. M. Gwilliam, S. Ledain, G. Shao, K. P. Homewood, Nature 410 (2001) 192-194.
- [4] O. V. Feklisova, E. B. Yakimov, N. A. Yarykin, J. Weber, Mater. Sci. Eng. B 58 (1999) 60-63.
- [5] Y. Ishibashi, T. Kobayashi, A. D. Prins, J. Nakahara, M. A. Lourenco, R. M. Gwilliam, K. P. Homewood, Phys. Status Solidi B 244 (2007) 402-406.
- [6] A. Podolyan, V. Khivrich, Tech. Phys. Lett. 31 (2005) 408-410.
- [7] O. Olikh, I. Ostrovskii, Phys. Solid State 44 (2002) 1249-1253.
- [8] I. Ostrovskii, N. Ostrovskaya, O. Korotchenkov, J. Reidy, IEEE Trans. Nucl. Sci. 52 (2005) 3068-3073.
- [9] A. Romanyuk, P. Oelhafen, R. Kurps, V. Melnik, Appl. Phys. Lett. 90 (2007) 013118-013113.
- [10] V. P. Melnik, Y. M. Olikh, V. G. Popov, B. M. Romanyuk, Y. Goltvyanskii, A. A. Evtukh, Mater. Sci. Eng. B 124 (2005) 327-330.
- [11] M. B. Tagaev, Funct. Mater. 5 (1998) 601-603.
- [12] Gromashevskiy V. L., Derevenko A. P., Ismailov K. A., K. R. V., Fiz. khim. obr. mater. 6 (1996) 132-135.
- [13] M. B. Tagaev, Ukr. J. Phys. 45 (2000) 364-367.
- [14] M. B. Tagaev, Dokl. Akad. Nauk of Uzbekistan 4 (1998) 52-54.
- [15] E. B. Zaveryukhina, N. N. Zaveryukhina, L. N. Lezilova, B. N. Zaveryukhin, V. V. Volodarskii, R. A. Muminov, Techn. Phys. Lett. 31 (2005) 27-32.
- [16] Y. M. Olikh, M. D. Tymochko, A. P. Dolgolenko, Tech. Phys. Lett. 32 (2006) 586-589.
- [17] M. A. Lourenco, M. Milosavljevic, R. M. Gwilliam, K. P. Homewood, G. Shao, Appl. Phys. Lett. 87 (2005) 201105.
- [18] M. A. Lourenco, M. Milosavljevic, S. Galata, M. S. A. Siddiqui, G. Shao, R. M. Gwilliam, K. P. Homewood, Vacuum 78 (2005) 551-556.
- [19] R. Gwilliam, M. A. Lourenco, M. Milosavljevic, K. P. Homewood, G. Shao, Mater. Sci. Eng. B 124 (2005) 86-92.
- [20] S. F. Galata, M. A. Lourenço, R. M. Gwilliam, K. P. Homewood, Mater. Sci. Eng.: B 124-125 (2005) 435.
- [21] P. L. Bradfield, T. G. Brown, D. G. Hall, Appl. Phys. Lett. 55 (1989) 100-102.
- [22] T. G. Brown, P. L. Bradfield, D. G. Hall, Appl. Phys. Lett. 51 (1987) 1585-1587.
- [23] T. G. Brown, D. G. Hall, Appl. Phys. Lett. 49 (1986) 245-247.
- [24] V. Kveder, M. Kittler, W. Schröter, Phys. Rev. B 63 (2001) 115208.
- [25] S. Z. Karazhanov, A. Davletova, A. Ulyashin, J. Appl. Phys. 104 (2008) 024501.
- [26] S. Z. Karazhanov, J. Appl. Phys. 89 (2001) 4030-4036.
- [27] A. Davletova, S. Z. Karazhanov J. Phys. D: Appl. Phys. 41 (2008) 165107.

Table I. Sulphur implantation dose D (cm<sup>-2</sup>), annealing temperature T (°C) to activate the sulphur atoms, UST power P and time  $\tau_{\mathit{UST}}$ , ideality coefficient  $\beta$  and saturation current  $J_0$  before and after UST.

Sampl	D,	<i>T</i> , (°C)	<i>P</i> ,	$ au_{\mathit{UST}}$ ,	β		$J_0 (\mu \text{A/cm}^2)$	
es	(cm <sup>-2</sup> )		(W/cm <sup>2</sup> )	(min)	Before	After	Before	After
					UST	UST	UST	UST
B1	10 <sup>13</sup>	1100	1.0	15	2.4	3.1	3.5	15.2
В3			0.5	30	2.3	2.3	3.3	3.6
B4			0.5	15	3.4	3.8	24.2	47.3
A1	10 <sup>14</sup>	1000	1.0	15	2.9	3.0	10.3	18.0
A2			1.0	30	2.8	2.8	8.0	9.5
A3			0.5	30	3.5	3.3	19.8	16.9
A4			0.5	15	2.3	2.5	2.6	3.5
C1	0	0	1.0	15	2.00	2.1	0.6	0.7
C2			1.0	30	1.8	2.0	0.4	0.5
C3			0.5	15	2.5	2.4	3.3	2.8
C4			0.5	30	1.9	2.0	0.9	0.5

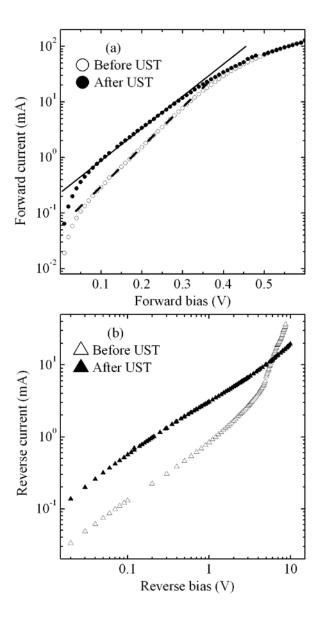


Fig. 1. Experimentally measured dark current-voltage characteristics for sample B1 for (a) forward and (b) reverse biases before  $(\circ, \Delta)$  and after  $(\bullet, \blacktriangle)$  UST. (a) The solid and dashed straight lines indicate that the J-V dependence can be described by linear relationship, i.e. by the Eq. (1). (b) For convenience of analysis the values of J and V have been multiplied to (-1) and the figure has been plotted in the first quadrant.

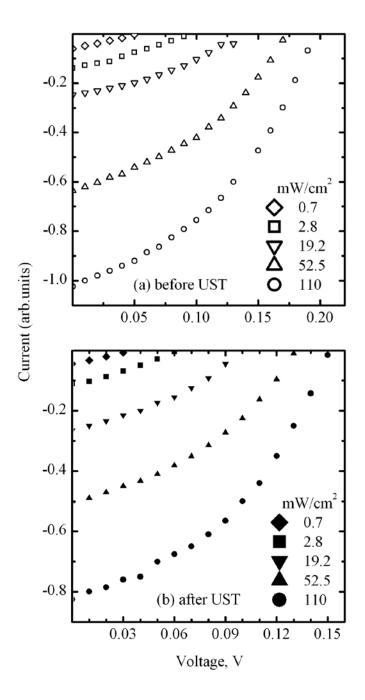


Fig. 2. Current-voltage characteristics for sample C2 (a) before and (b) after UST for different illumination intensities.

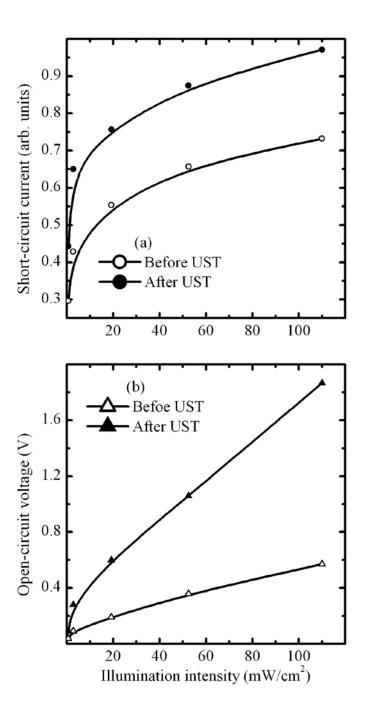


Fig. 3. Dependence of (a) photocurrent and (b) photovoltage on illumination intensity for the sample A1 before and after UST.