

Phase stability and pressure-induced structural transitions at zero temperature in ZnSiO_3 and Zn_2SiO_4

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Abstract

Using density functional total energy calculations the structural phase stability and pressure-induced structural transition in different polymorphs of ZnSiO_3 and Zn_2SiO_4 have been studied. Among the considered monoclinic phase with space groups $P2_1/c$ and $C2/c$, rhombohedral ($R\bar{3}$), and orthorhombic ($Pbca$) modifications the monoclinic phase ($P2_1/c$) of ZnSiO_3 is found to be the most stable one. At high pressure monoclinic ZnSiO_3 ($C2/c$) can coexist with orthorhombic ($Pbca$) modification. Difference in equilibrium volume and total energy of these two polymorphs are very small, which indicates that it is relatively easier to transform between these two phases by temperature, pressure or chemical compositions. It can also explain the experimentally established result of metastability of the orthorhombic phase under all conditions. The following sequence of pressure induced structural phase transitions are found for ZnSiO_3 polymorphs: monoclinic ($P2_1/c$) \rightarrow monoclinic ($C2/c$) \rightarrow rhombohedral ($R\bar{3}$). Among the rhombohedral ($R\bar{3}$), tetragonal ($I\bar{4}2d$), orthorhombic ($Pbca$), orthorhombic ($Imma$), cubic ($Fd\bar{3}m$), and orthorhombic ($Pbnm$) modifications of Zn_2SiO_4 , rhombohedral phase is found to be the ground state. For this chemical composition of the zinc silicate the following sequence of structural phase transitions is found: rhombohedral ($R\bar{3}$) \rightarrow tetragonal ($I\bar{4}2d$) \rightarrow orthorhombic ($Pbca$) \rightarrow orthorhombic ($Imma$) \rightarrow cubic ($Fd\bar{3}m$) \rightarrow orthorhombic ($Pbnm$). Based on the analogy of crystal structures of magnesium and zinc silicates and using the lattice and positional parameters of Mg_2SiO_4 as input, structural properties of spinel Zn_2SiO_4 have also been studied.

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1. Introduction

Studies of the interface between semiconductors and transparent conducting oxides (TCO) present much interest for optoelectronic device technology. During fabrication of device structures based on contact semiconductor/TCO a mutual diffusion of the atoms from semiconductor to TCO and vice versa can take place at the interface. At high concentrations of the diffused atoms structural phase transformations can occur, which can result in formation of new compounds at the interface. Such compounds often possess wide band gap and form band offset with the semiconductor and TCO. From this point of view knowledge of structural properties, electronic structure, and optical properties of the compounds formed at the interface is important to use them in optoelectronic devices. Further, the crystal structure of compounds formed at the interface will be different from that of the bulk materials due to the interfacial effect and stress in the lattice. So, the structural phase stability studies on such compounds with pressure are both fundamental and technological significance. It can help to reveal how the interface influences operation of a particular device and to provide the information required to control the properties of the interface layer.

ZnO and Si are the compounds extensively used in modern semiconductor electronics. Electronegativity of Si and Zn are 1.90 and 1.65, respectively, which are close to each other. The atomic sizes of these elements are also comparable to each other i.e. 0.117 nm for Si and 0.133 nm for Zn. These data indicate that the diffusion of Si into ZnO can be favorable. As a result, at the interface in the heterostructures of Si/ZnO and SiO₂/ZnO the Si and Zn related oxides can be formed, which will affect the device performance. Polymorphs of ZnSiO₃ and Zn₂SiO₄ can be such interfacial oxide layers formed in between Si/ZnO and SiO₂/ZnO[(1-2)] due to the solid state reaction. These compounds are of particular technological interest for thin film Si solar cells,[(3-6)] devices where ZnO deposited on silicate glass as a TCO buffer layer,[(7)] light-emitting diodes,[(8)] ZnO nanowires and nanotips grown on Si and Si on sapphire,[(9)] ZnO particles embedded in sol-gel silica,[(1)] nanostructures, etc..[(10-16)] The monoclinic ZnSiO₃ nanocrystals have been formed as a result of rapid thermal annealing of SiO₂ with metallic Zn nanocrystals.[(17)] Also by the transmission electron microscopy (TEM) with a focused electron beam irradiation process ZnSiO₃ nanoparticles have been found in a SiO₂ layer located in between the ZnO thin film and the Si substrate.[(18)] Moreover, from x-ray diffraction, TEM, and selected-area

electron diffraction studies it has been found that orthorhombic ZnSiO_3 can be formed at the interface of the ZnO/Si heterostructure after annealing at 900 °C resulting from the inter-diffusion between ZnO and Si. [(19)]

Crystal structure and structural phase transitions of ZnSiO_3 and Zn_2SiO_4 have been studied experimentally in the past by many authors (see, e.g., Ref. [(20-25)]) at high temperatures in the range 700-1500 °C and pressures 0-17 GPa. ZnSiO_3 in the monoclinic phase with space group (SG) $C2/c$ is generally not found to be stable at atmospheric pressure and it can be obtained as a result of the reaction of willemite Zn_2SiO_4 with quartz at about 3 GPa and subsequent transition to ilmenite at 10-12 GPa. [(20-25)] or 15 GPa. [(26)] However, it is reported in Ref. [(21)] that ZnSiO_3 with SG $C2/c$ exist at room temperature and ambient conditions also. Upon increasing the pressure, this phase can undergo the following sequence of displacive phase transitions from $C2/c$ to a high pressure phase with SG $P2_1/c (-m_1)$ at 1.92 GPa, which can also be transformed to another high pressure phase with SG $C2/c (-m_2)$ at 4.9 GPa. However, the differences in equilibrium volumes between the latter three structural modifications of ZnSiO_3 are too small to consider them as different polymorphs. Orthorhombic ($-o$; $Pbca$) ZnSiO_3 has been synthesized at high pressures and temperatures (see, e.g., Ref. [(24)]) and it is found to be metastable under all conditions.

In the polymorphs of ZnSiO_3 , Zn atoms are coordinated both octahedrally and tetrahedrally [(24)] (Fig. 1). In $\text{ZnSiO}_3\text{-}o$ the Zn atoms at the octahedral sites have irregular octahedral coordination to the O atoms. ZnSiO_3 of trigonal/rhombohedral ($-r$) symmetry with space group $R\bar{3}$ is known by mineral name ilmenite. $\text{ZnSiO}_3\text{-}r$ consists of a slightly distorted hexagonal close-packing array of O atoms with Zn and Si atoms in the interstices. Structural properties of these compounds have been analyzed more in detail in Refs. [(20, 23, 27)]. However, phase stability of different modifications of ZnSiO_3 at low temperatures is not yet discussed in detail. According to Ref. [(26)] structural transition from $\text{ZnSiO}_3\text{-}m_2$ to $-r$ occurs at 15 GPa at 1000 °C.

The Zn_2SiO_4 reported in Ref. [(28)] has been formed as a result of annealing the ZnO-SiO_x system., Willemite Zn_2SiO_4 has been formed as a result of the reaction of ZnO with SiO₂, [(29)] which has lead to decrease of concentration of the O vacancies and Zn interstitials. Zn_2SiO_4 has been used to synthesize the Si doped ZnO nanobelts. [(30)]

Phase transformations of Zn_2SiO_4 have been studied[(31, 24, 27)] up to pressures ≤ 17 GPa and temperatures $800\text{ }^\circ\text{C} - 1500\text{ }^\circ\text{C}$. The following sequence of structural transformations have been found[(27)] between five polymorphs of Zn_2SiO_4 : rhombohedral(r) \rightarrow tetragonal(t) \rightarrow monoclinic(m) \rightarrow orthorhombic(o_1) \rightarrow orthorhombic(o_2) \rightarrow orthorhombic(o_3). Here the o_1 , o_2 , and o_3 modifications of the orthorhombic Zn_2SiO_4 denote those with SG symmetry $Pbca$, $Imma$, and $Pbnm$, respectively. Owing to the preference of Zn for tetrahedral rather than octahedral coordination by oxygen at low pressures, the Zn_2SiO_4 crystallizes into the phenacite-structure with Zn atoms located in distorted octahedral coordination. [(14, 26, 22-23)] In Zn_2SiO_4 - t the oxygen atoms join together to Si and Zn so that each of the oxygen at the SiO_4 tetrahedra is bonded to Zn atoms thus forming an integrated part of a Zn-O-Si network (see Fig. 2). The O atoms are located approximately in the body-centered cubic arrangement in Zn_2SiO_4 - t , while both the Zn and Si atoms are in tetrahedral coordination to the O atoms. In rhombohedral Zn_2SiO_4 also the Zn and Si atoms are tetrahedrally coordinated to O atoms, where the Zn atoms occupy two crystallographically different sites, but the environments around these sites differ only slightly. Although there is no information about spinel structure of Zn_2SiO_4 , similar type of compound Mg_2SiO_4 posses it. The spinels are cubic ($-c$). They are a class of minerals with the O ions forming a face-centered lattice where Zn cations are octahedrally coordinated and Si atoms are tetrahedrally coordinated.

The above mentioned experimental results about existence of different polymorphs of the zinc silicates, possibility of their formation at the interface between ZnO-Si and ZnO-SiO₂, and possibility of phase transitions between the polymorphs creates the necessity to study structural properties, electronic structure, optical properties, as well as interface between the silicates and semiconductors. The study about electronic structure and optical properties of the intermixed zinc silicates has been performed in Ref. [(32)] which are found to be insensitive to crystal structure modifications and crystallographic directions. Still many questions have been left open. Some of them are the systematic study of band gap variation in between the zinc silicates and ZnO(Si) and the ways of controlling them, which would be important to know influence of the interface layer on current transport. The other one is doping of the silicates by shallow level impurities and H, which allows one to classify the silicates as semiconductors or insulators. However, first of all it is interesting to know which of the polymorphs of the zinc silicates is the most stable one, whether structural phase transitions can take place at low

temperatures and pressures as well and what is difference between equilibrium volumes of the polymorphs. In the present paper we intend to study these questions for ZnSiO_3 *-o*, *-m₁*, *-m₂*, and *-r* as well as Zn_2SiO_4 *-t*, *-r*, *-m*, *-o₁*, *-o₂*, *-o₃*, *-c*.

2. Methods

The computations have been performed using the density-functional theory (DFT) within the local-density approximation (LDA). Structural relaxations and calculation of total energies have been performed with the projector-augmented-wave method (PAW) [(33-34)] as implemented in the Vienna *ab initio* simulation package (VASP)[(35)] using the Ceperley-Alder data[(36)] for the correlation energy in the parameterization by Perdew-Zunger.[(37)] The Zn-3*d*, and -4*s*, O -2*s* and -2*p*, Si-3*s* and -3*p* have been considered as the valence electrons. The self-consistent calculations were performed using a 4×4×4 mesh of special **k**-points. Test calculations showed that plane-wave cutoff energy of 500 eV is sufficient to describe structural properties of zinc silicates reliably. The convergence was achieved when the forces acting on the atoms were <10 meV Å⁻¹ and the total energy difference between two consecutive iterations were < 10⁻⁶ eV.

The crystal structures are of different modifications of ZnSiO_3 and Zn_2SiO_4 are presented in Figs. 1 and 2, respectively. Experimentally determined lattice parameters and positional parameters[(38, 21)] have been used as input. All configurations were fully relaxed using the conjugate gradient method. Here it should be noted that there is some similarity between structural phase transitions and crystal structures of ZnSiO_3 and MgSiO_3 .[(20, 23-25)] The latter compound is well studied and is very important material to explore the mineralogy of the Earth's deep mantle. So, MgSiO_3 can be used as a model system in the structural studies of ZnSiO_3 . In the present paper this similarity shall be used. Due to lack of experimental data for positional and lattice parameters of ZnSiO_3 -*r* and Zn_2SiO_4 -*c* we have used those for MgSiO_3 -*r* and Mg_2SiO_4 -*c*, respectively as input for the structural optimization for these two phases.

3. Results

3.1 ZnSiO₃

Structural optimization has been performed for all the considered structural modifications by the following procedure: first, atomic positions have been relaxed by force minimization method keeping the volume and shape of the crystal fixed. Using the relaxed atomic positions as input, the crystal volume and shape have been relaxed using stress minimization freezing out atomic positions. Then these optimized parameters have been used as input to relax atomic positions, cell volume and shape altogether. Crystal structure information obtained by this way was used as input for calculation of the total energy (E_{tot}) as a function of the cell volume (V). The minima ($E_{\text{tot}}^{\text{min}}$) of the dependence $E_{\text{tot}}(V)$ are taken as the equilibrium volume. Positional and lattice parameters derived from the DFT calculations for the equilibrium lattices are given in Table I together with experimentally determined values. Analysis of the Table I show that deviation of the calculated equilibrium volumes from the corresponding experimentally determined values by 2.3% for ZnSiO_{3-*o*}, 3.3% for ZnSiO_{3-*m*₁}, and 2.4% for ZnSiO_{3-*m*₂}. The calculated positional parameters from the structural optimization are found to be overall in good agreement with experimental data. Symmetry of the optimized lattices were checked and found that it is the same as that of the experimentally determined one used as an input for the structural optimization.

Dependence of the total energy on volume has been analyzed for ZnSiO₃ (Fig. 3). The $E_{\text{tot}}(V)$ for ZnSiO_{3-*o*} is 2.3 meV higher in energy than ZnSiO_{3-*m*₂} at their total energy minimum. Magnitude of the $E_{\text{tot}}^{\text{min}}$ for ZnSiO_{3-*m*₁} is found to be lower than that of ZnSiO_{3-*m*₂} by about 10 meV. Consequently, m_2 is the most stable phase. $E_{\text{tot}}^{\text{min}}$ for ZnSiO_{3-*r*} is found to be about 10 meV higher than that of ZnSiO_{3-*o*} which indicates that ZnSiO_{3-*r*} is the least stable among the considered phases. Since the difference in $E_{\text{tot}}(V)$ between ZnSiO_{3-*o*}, ZnSiO_{3-*m*₁} and $-m_2$ is not large, a small fluctuations of temperature and pressure can be sufficient to cause the phase transformation between these phases. This result could explain why metastability is observed experimentally for ZnSiO_{3-*o*} at almost all conditions. [(24)]

Upon compression the volume dependence of E_{tot} for ZnSiO_{3-*m*₁} intersects with that for $-m_2$, $-r$, and $-o$ (Fig. 3), which indicates the possibility of phase transition between these modifications. The present calculations suggest that upon compression the following sequence of structural phase transitions can take place: one from

ZnSiO_{3-*m*1} to ZnSiO_{3-*m*2} and the other from ZnSiO_{3-*m*2} to ZnSiO_{3-*r*} at pressures 6.8 and 8.5 GPa, respectively, with corresponding volume shrinkage at the phase transition point 1.1 and 5.24 Å³ per formula unit. The calculated transition pressure of 6.8 GPa for the transition ZnSiO_{3-*m*1} → ZnSiO_{3-*m*2} somewhat agrees with that of 4.9 GPa obtained from experimental high pressure measurements.[(21)] The calculated transition pressure for the ZnSiO_{3-*m*1} → ZnSiO_{3-*r*} phase transition is 11.0 GPa and this agrees well with the experimental observation of pressure induced phase transition from ZnSiO_{3-*m*} to -*r* at 10-12 GPa[(20-24)] and smaller than 15 GPa.[(26)] Furthermore, our studies of the pressure dependence of the volume for ZnSiO_{3-*r*} (Fig. 5) is found to be in good agreement with recent experimental data from Ref.[(25)].

Bulk modulus (B_0) is the parameter characterizing compressibility of a solid. Among the considered polymorphs of ZnSiO₃, the calculated B_0 values vary in the range from 47.4 GPa for ZnSiO_{3-*m*2} to 235.8 GPa for ZnSiO_{3-*o*} (Table I). These analysis shows that ZnSiO_{3-*o*} is the hardest phase among the considered polymorphs. The pressure derivative of the bulk modulus (B_0') for some of the silicates is surprisingly large. It equals to 4.58 for ZnSiO_{3-*o*}, 7.15 for ZnSiO_{3-*m*1}, and 5.5 for ZnSiO_{3-*r*}. We have cross checked the calculated bulk modulus and its pressure derivative with two independent fitting programs as well as with different equation of state models and arrived at the same results.

3.2 Zn₂SiO₄

From the structural optimization the equilibrium structural parameters for different polymorphs of Zn₂SiO₄ have been calculated and presented in Table II. Analysis of the Table II shows that the calculated equilibrium volumes differ from the experimentally determined values by <1.4% for Zn₂SiO_{4-*o*2}, 3.2% for Zn₂SiO_{4-*t*} and 3.4% for Zn₂SiO_{4-*r*}. For Zn₂SiO_{4-*m*} the calculated equilibrium volume is considerable smaller by 8.6% from the experimentally determined one. The discrepancy between the calculated and experimental equilibrium volume for different polymorphs may be related to the difference in stoichiometry of the experimentally studied phases, temperature effects, and also uncertainty in some of the lattice and positional parameters. For example, the lattice parameters for Zn₂SiO_{4-*m*} and Zn₂SiO_{4-*o*1} in Ref. [(27)] correspond to non-stoichiometric crystals with atomic ratio of Zn:Si=1.7 and 1.9, respectively, whereas in our calculations the

atomic ratio was always Zn:Si=2. Overall the calculated positional parameters agree well with experimental data given in Table II. We have checked the symmetry of the lattices after structural relaxation and found that their symmetry remain the same as that of the experimentally determined one for all considered polymorphs.

Dependence of the total energy with volume for Zn_2SiO_4 -*r*, -*t*, -*o*₁, -*o*₂, -*c*, -*o*₃ are displayed in Fig. 6. Among these polymorphs, Zn_2SiO_4 -*r* possesses the largest equilibrium volume and smallest total energy at the minimum. Zn_2SiO_4 -*c* posses the smallest equilibrium volume, but Zn_2SiO_4 -*m* has largest total energy at $E_{\text{tot}}^{\text{min}}$ among the considered phases. Consequently, according to our calculations Zn_2SiO_4 -*r* is the most stable phase while Zn_2SiO_4 -*m* is the least stable phase. In addition to the largest total energy, the dependence $E_{\text{tot}}(V)$ for Zn_2SiO_4 -*m* is crossed by that of Zn_2SiO_4 -*o*₁ at its minimum. Consequently, even if Zn_2SiO_4 -*m* has been formed, it can be transformed into Zn_2SiO_4 -*o*₁, without pressing or heating.

Analysis of Fig. 6 shows that the total energy variation with volume for Zn_2SiO_4 -*r* intersects with Zn_2SiO_4 -*t* at high pressure. At more volume compression, $E_{\text{tot}}(V)$ for other polymorphs also intersect in the following sequence: Zn_2SiO_4 -*t* with -*o*₁, -*o*₁ with -*o*₂, -*o*₂ with -*c*, and -*c* with -*o*₃. This result indicates the possibility of pressure-induced phase transition, which is demonstrated in Fig. 7. The phase transition from Zn_2SiO_4 -*r* to Zn_2SiO_4 -*t* occurs at 0.8 GPa with volume shrinkage 7.3 Å³/f.u., whereas Zn_2SiO_4 -*t* to Zn_2SiO_4 -*o*₁ takes place at 15.0 GPa with very small volume shrinkage 0.8 Å³/f.u.. Zn_2SiO_4 -*o*₁ can exist at small pressure range and upon compression it can be transformed into Zn_2SiO_4 -*o*₂. The volume variation at the transition Zn_2SiO_4 -*o*₁ to -*o*₂ occurring at 15.8 GPa, is ~4.7 Å³/f.u.. Upon compression Zn_2SiO_4 -*o*₂ can transform into Zn_2SiO_4 -*c* at 20.0 GPa (Fig. 7) with volume shrinkage 2.1 Å³/f.u. Zn_2SiO_4 -*c* can be transformed to Zn_2SiO_4 -*o*₃ at 24.9 GPa with small volume variation ~1.3 Å³/f.u. It should be noted that the present calculations are strictly speaking valid only for the stoichiometric compounds at low temperatures. In contrast off-stoichiometry and temperature could stabilize the experimentally identified high pressure polymorphs such as Zn_2SiO_4 -*m*, -*o*₁, -*o*₂, and -*o*₃. In order to clarify the origin of stability of these polymorphs at high pressures and low temperature high pressure experimental measurements on stoichiometric Zn_2SiO_4 at low temperatures are needed.

The calculated bulk modulus for Zn_2SiO_4 polymorphs (Table II) vary in the range from 63.0 GPa for Zn_2SiO_4 -*m* to 235.8 GPa for Zn_2SiO_4 -*o*₃. It follows from these analyses that Zn_2SiO_4 -*o*₃ is the hardly

compressible phase. The calculated bulk modulus and its pressure derivative for Zn_2SiO_4-c is found to be in good agreement (see Table II) with the respective experimentally determined values of 212 GPa and $B_0' = 4$ for Mg_2SiO_4-c [(39)] and the bulk modulus of 216 GPa for Zn_2SiO_4-c [(23)]. This result indicates that the valence and bonding behavior of Zn_2SiO_4 is similar to that of the geochemically important material Mg_2SiO_4 . However, among the considered Zn_2SiO_4 polymorphs, Zn_2SiO_4-m is found to be the softest phase.

4. Discussion

One of the important questions is which of the $ZnSiO_3$ and Zn_2SiO_4 polymorphs can be formed at the interface $ZnO-Si(SiO_2)$ and how it depends on growth and device operation conditions. Although some preliminary studies are already available in literature, there is no systematic study of the point. For example, a crystalline $ZnSiO_3$ is formed[(40)] upon irradiation of nanocomposite $ZnO-SiO_2$ films with ultraviolet light because of the photo-induced reaction. From analysis of secondary ion mass spectroscopy inter-diffusion of $Zn(Si)$ into $Si(ZnO)$ has been reported at the interface between ZnO film and Si substrate.[(2)] By means of cathodoluminescence and glancing-angle X-ray diffraction the tetragonal modification of Zn_2SiO_4 was proved to be present. Zn_2SiO_4 is also formed at the interface of ZnO/Si heterostructures.[(5, 2, 41)] It can form large band offset with Si and ZnO ,[(5)] which is assumed to be one of the reasons for low 8.5 % efficiency of crystalline $Si-ZnO$ solar cells.[(3)] Formation of rhombohedral modification of Zn_2SiO_4 at the boundary between ZnO particles and SiO_2 matrix has also been reported.[(16)] These findings are important and indicate that the idea about formation of the zinc silicates is not far from reality. From the theoretical study of the present paper we found that $ZnSiO_3-m_1$ and Zn_2SiO_4-r are the most stable polymorphs. Systematic experimental study of this point is needed.

The other important point is the pressure at which the stable polymorphs $ZnSiO_3-m_1$ and Zn_2SiO_4-r can be transformed into another one. Analysis of the above theoretical findings corresponding to $T=0$ K and experimental results corresponding to high temperatures indicate that the transition pressure is very large, which might not be available in $ZnSiO_3$ or Zn_2SiO_4 containing device structures. Consequently, possibility of phase transitions at ambient pressures (<6.82 GPa for $ZnSiO_3$ and <0.80 GPa for Zn_2SiO_4) can be excluded. This

finding has important implications such as, for example, the possibility of lattice expansion/shrinkage, phase-transition-induced abnormal temperature dependence of electrical conductivities experimentally observed[(42)] in ZnTe:Cu can also be excluded.

Comparative analysis[(32)] of electronic structure and optical properties of the ZnSiO₃ and Zn₂SiO₄ shows that optical parameters and conduction band electron effective masses of the compounds are almost isotropic and their dependence on crystal structure is negligible. Furthermore, the calculated absorption coefficient, reflectivity, refractive index, and extinction coefficients are $<10^3 \text{ cm}^{-1}$, 0.15, 2.2, and 0.3, respectively, which indicate the possibility of using the zinc silicates as antireflection coatings. Consequently, pressure-induced phase transitions is not expected to cause modulation of optical properties of the ZnO-ZnSiO₃(Zn₂SiO₄)-Si based device structures.

5. Conclusion

Phase stability and pressure-induced structural transition between different polymorphs of ZnSiO₃ and Zn₂SiO₄ have been studied based on *ab initio* total energy calculations. Among the monoclinic phase with space groups $P2_1/c$ and $C2/c$, rhombohedral ($\bar{R}3$), and orthorhombic ($Pbca$) modifications, the low pressure monoclinic ZnSiO₃ phase is found to be the most stable one. Two sequences of structural phase transitions between different modifications of the ZnSiO₃ have been found. One sequence is from ZnSiO₃ ($P2_1/c$) to ZnSiO₃-($C2/c$) and the other one is from ZnSiO₃-($C2/c$) to ZnSiO₃- $\bar{R}3$. At high pressure monoclinic ZnSiO₃ ($C2/c$) can coexist with orthorhombic ($Pbca$) and equilibrium volume and total energy corresponding to these two polymorphs are found to be slightly different from each other. This indicates that small fluctuation in environment can be sufficient for orthorhombic to monoclinic transition, which could also explain why the orthorhombic phase is metastable under all conditions. These two polymorphs are found to co-exist in a certain pressure and volume range. According to present total energy calculations ilmenite form of ZnSiO₃ is found to be the most unstable one at low temperatures and high pressures. Distinct from previous experimental studies at high temperatures, no pressure-induced phase transition has been found into/from this modification. This

indicates that the ilmenite phase of ZnSiO_3 may be stabilized by the lattice dynamics or oxygen stoichiometry. Among the rhombohedral, tetragonal, orthorhombic, orthorhombic, cubic, and orthorhombic modifications of Zn_2SiO_4 rhombohedral phase is found to be most stable. It can be transformed into tetragonal phase at high pressures, which upon further compression transformed into spinel modification. The calculated transition pressures agree with experimental data. Based on analogy with Mg_2SiO_4 orthorhombic Zn_2SiO_4 has been studied. It is found that this polymorph possesses relatively large bulk modulus and, consequently, can be considered as hardly compressible phase. The presently considered silicates can exist at the interface between ZnO and $\text{Si}(\text{SiO}_2)$. Although the reported structural transformation between the zinc silicate polymorphs is possible, the transitions occur at high pressures. So, at low pressures, the phase transition-induced lattice expansion/shrinkage or modulation of the electronic structure and optical properties of the device structures with ZnSiO_3 and Zn_2SiO_4 is not expected.

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Table I Unit-cell dimensions and volumes, positional parameters, bulk modulus (B_0), as well as the derivative of bulk modulus (B_0') for ZnSiO_3 , $-o$, $-m_1$, $-m_2$, $-m_3$, $-m_4$, $-r_1$, and $-r_2$ derived from the density functional total energy calculations at the total energy minimum. Values given in parentheses refer to experimental data. [(38)] The unit cell volumes are presented per formula unit in \AA^3 .

Compound	Unit cell (\AA)	Atom	Site	X	Y	Z	B_0 (GPa)	B_0'
ZnSiO_3 - o	$a=17.4224(18.2040)^a$	Zn1	$8c$	0.1253 (0.1255)	0.3557 (0.3559)	0.4007 (0.4039)	235.8	4.58
$Pbca$	$b=8.7544(9.0870)^a$	Zn2	$8c$	0.3760 (0.3761)	0.5140 (0.5092)	0.3949 (0.4005)		
	$c=5.0881(5.2780)^a$	Si1	$8c$	0.2742 (0.2741)	0.3373 (0.3363)	0.0852 (0.0867)		
	$V=53.32(54.57)^a$	Si2	$8c$	0.4717 (0.4730)	0.1677 (0.1656)	0.2872 (0.2839)		
		O1	$8c$	0.1840 (0.1853)	0.3318 (0.3346)	0.0831 (0.0820)		
		O2	$8c$	0.3108 (0.3115)	0.4995 (0.4966)	0.1045 (0.1040)		
		O3	$8c$	0.3032 (0.3039)	0.2503 (0.2521)	0.3368 (0.3400)		
		O4	$8c$	0.5624 (0.5621)	0.3355 (0.3389)	0.7863 (0.7800)		
		O5	$8c$	0.4301 (0.4327)	0.4811 (0.4853)	0.6999 (0.7040)		
		O6	$8c$	0.4473 (0.4488)	0.2049 (0.2100)	0.5805 (0.5710)		
ZnSiO_3 - m_1	$a=9.6543(9.5781)^b$	Zn1	$4e$	0.2494 (0.2508)	0.6465 (0.6509)	0.2332 (0.2247)	66.5	7.15
$P2_1/c$	$b=9.0667(8.8905)^b$	Zn2	$4e$	0.2530 (0.2551)	0.0129 (0.0034)	0.2312 (0.2211)		
	$c=5.2653(5.1798)^b$	Si1	$4e$	0.0507 (0.0468)	0.3370 (0.3403)	0.2906 (0.2924)		
	$V=53.69(51.99)^b$	Si2	$4e$	0.5583 (0.5568)	0.8332 (0.8338)	0.2500 (0.2397)		
	$\beta=111.26(109.443)^b$	O1	$4e$	0.8693 (0.8691)	0.3323 (0.3386)	0.1652 (0.1800)		
		O2	$4e$	0.3764 (0.3779)	0.8362 (0.8367)	0.1309 (0.1300)		
		O3	$4e$	0.1231 (0.1169)	0.5000 (0.5033)	0.3537 (0.3365)		
		O4	$4e$	0.6382 (0.6341)	0.9845 (0.9828)	0.3906 (0.3895)		
		O5	$4e$	0.1119 (0.1090)	0.2529 (0.2662)	0.5850 (0.5982)		
		O6	$4e$	0.6079 (0.6078)	0.7051 (0.6965)	0.4893 (0.4702)		
ZnSiO_3 - m_2	$a=9.3173(9.787)^a$	Zn1	$4e$	0.5000 (0.5000)	0.3921 (0.3919)	0.2500 (0.2500)	47.4	4.2
$C2/c$	$b=9.8180(9.161)^a$	Zn2	$4e$	0.5000 (0.0000)	0.7289 (0.2361)	0.2500 (0.2500)	(72.0) ^b	(7.0) ^b
	$c=5.0898(5.296)^a$	Si1	$8f$	0.3060 (0.3016)	0.0837 (0.0849)	0.2646 (0.2668)	(101.0)	
	$V=53.92(55.26)^a$	O1	$8f$	0.1238 (0.1241)	0.0833 (0.0868)	0.1427 (0.1473)		
	$\beta=111.80(111.42)^a$	O2	$8f$	0.3839 (0.3787)	0.2390 (0.2393)	0.3783 (0.3719)		
		O3	$8f$	0.3583 (0.3533)	0.0321 (0.0238)	0.0184 (0.0273)		
ZnSiO_3 - r	$a=4.8115(4.7469)^c$	Zn	$6c$	0.0000 (0.0000)	0.0000 (0.0000)	0.3692 (0.3599)	177.89	5.5
$R\bar{3}$	$c=13.9974(13.754)^c$	Si	$6c$	0.0000 (0.0000)	0.0000 (0.0000)	0.1607 (0.1556)	(216.0) ^{c,d}	
	$V=46.77(44.73)^c$	O	$18f$	0.9911 (0.9645)	0.3736 (0.3200)	0.4286 (0.4896)		

^aFrom Ref. [(38)].

^bFrom Ref. [(21)].

^cFrom Ref. [(23)].

^dFrom Ref. [(23, 25)].

Table II Unit-cell dimensions, equilibrium volumes, positional parameters, bulk modulus (B_0), and the pressure derivative of bulk modulus (B_0') for Zn_2SiO_4 $-t$, $-r$, $-m$, $-o_1$, $-o_2$, and $-c$ derived from the density functional total energy calculations at the total energy minimum. Values given in parentheses refer to experimental data. [(38)]

Compound	Unit cell (Å)	Atom	Site	X	Y	Z	B_0 (GPa)	B_0'
Zn_2SiO_4-r	$a=13.8469(13.971)^a$	Zn1	$18f$	0.1908 (0.1916)	0.2098 (0.2087)	0.5843 (0.5844)	138.8	3.3
$R\bar{3}$	$c=9.1781(9.3340)^a$	Zn2	$18f$	0.5264 (0.5579)	0.8818 (0.9655)	0.5841 (0.5815)		
	$V=84.670(87.66)^a$	Si	$18f$	0.1951 (0.1963)	0.2127 (0.2118)	0.2499 (0.2510)		
		O1	$18f$	0.7924 (0.8767)	0.5435 (0.5415)	0.4376 (0.3918)		
		O2	$18f$	0.1271 (0.1295)	0.2063 (0.3178)	0.3974 (0.3955)		
		O3	$18f$	0.1083 (0.1100)	0.2127 (0.2175)	0.7507 (0.7496)		
		O4	$18f$	0.9854 (0.9335)	0.6568 (0.6283)	0.5838 (0.5007)		
Zn_2SiO_4-t	$a=6.9562(7.0069)^b$	Zn	$8d$	0.1549 (0.1570)	0.2500 (0.2500)	0.1250 (0.1250)	135.8	1.3
$I\bar{4}2d$	$c=6.3533(6.4637)^b$	Si	$4b$	0.0000 (0.0000)	0.0000 (0.0000)	0.5000 (0.5000)		
	$V=76.86(79.38)^b$	O	$16e$	0.3054 (0.3079)	0.4832 (0.4890)	0.1070 (0.1376)		
Zn_2SiO_4-m	$a=5.3752(5.069)^c$	Zn1	$4e$	0.3542	0.3811	0.5969	63.0	1.7
$P2_1/n$	$b=7.8794(10.292)^c$	Zn2	$4e$	0.4476	0.9915	0.3091		
	$c=7.5624(6.667)^c$	Si	$4e$	0.1760	0.7380	0.5414		
	$V=79.55(87.0)^c$	O1	$4e$	0.2697	0.9276	0.5053		
	$\beta=96.55(\sim 90)^c$	O2	$4e$	0.0135	0.7350	0.7089		
		O3	$4e$	0.4237	0.6246	0.5874		
		O4	$4e$	0.0019	0.6468	0.3805		
$Zn_2SiO_4-o_1$	$a=4.7231(4.79)^c$	Zn1	$4a$	0.0000	0.0000	0.0000	157.8	4.3
$Pbnm$	$b=10.1975(10.3)^c$	Zn2	$4c$	0.9850	0.2803	0.2500		
	$c=5.9430(6.02)^c$	Si	$4c$	0.4259	0.0960	0.2500		
	$V=71.56(74.25)^c$	O1	$4c$	0.7642	0.0936	0.2500		
		O2	$4c$	0.2164	0.4491	0.2500		
		O3	$8d$	0.2797	0.1643	0.0335		
$Zn_2SiO_4-o_2$	$a=5.8096(5.740)^c$	Zn1	$4a$	0.0000	0.0000	0.0000	222.0	11.1
$Imma$	$b=11.6392(11.504)^c$	Zn2	$4e$	0.2500	0.7500	0.4683		
	$c=8.5397(8.395)^c$	Zn3	$8g$	0.2500	0.1254	0.2500		
	$V=69.29(69.30)^c$	Si	$8h$	0.0000	0.1208	0.6164		
		O1	$4e$	0.0000	0.2500	0.2125		
		O2	$4e$	0.0000	0.2500	0.7130		
		O3	$8h$	0.0000	0.4873	0.7579		
		O4	$16j$	0.2639	0.1229	0.9917		
$Zn_2SiO_4-o_3$	$a=4.8891$	Zn	$8c$	0.9964	0.0553	0.3367	235.8	4.6
$Pbca$	$b=4.9980$	Si	$4a$	0.0000	0.0000	0.0000		
	$c=10.1666$	O1	$8c$	0.2012	0.2874	0.0494		
	$V=62.11$	O2	$8c$	0.8850	0.9553	0.1588		
Zn_2SiO_4-c	$a=8.0755^c$	Zn	$8b$	0.5000	0.5000	0.5000	222.2	4.6
$Fd\bar{3}m$	$V=65.83^c$	Si	$32e$	0.8750	0.8750	0.8750		
		O	$32e$	0.2438	0.2438	0.2438		

^a From Ref. [(43)].

^b From Ref. [(31)].

^c Only lattice parameters of Ref. [(27)]. have been used and the positional parameters are from Ref. [(38)] with similar chemical formula and SG, i.e. Ca_2SiO_4 type structure with SG $P2_1/n$ for Zn_2SiO_4-m , Ca_2RuO_4 type structure with SG $Pbca$ for $Zn_2SiO_4-o_3$, Mg_2SiO_4 type structure with SG $Imma$ for $Zn_2SiO_4-o_2$, Mg_2SiO_4 type structure with SG $Pbnm$ for $Zn_2SiO_4-o_1$ have been used for the present computations. Lattice and positional parameters for Zn_2SiO_4-c we have used Mg_2SiO_4 type structure with SG $Fd\bar{3}m$.

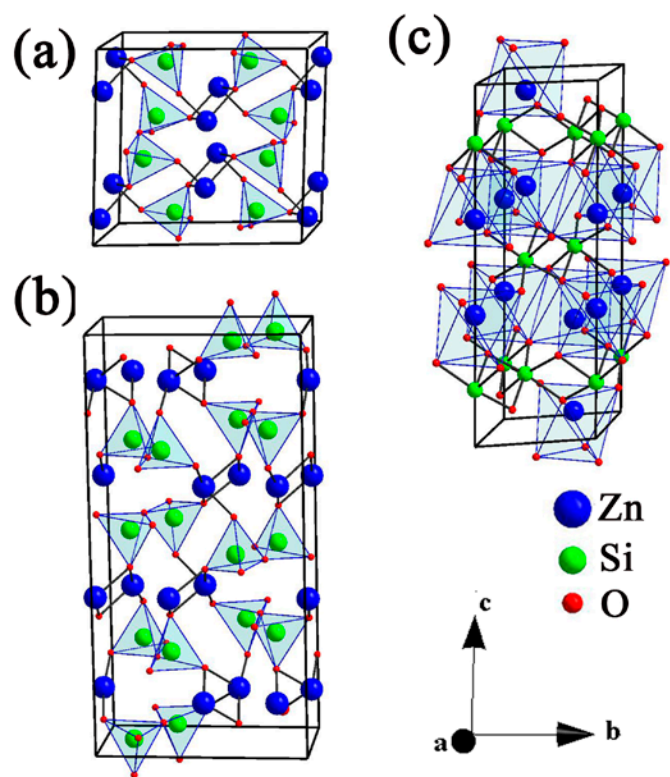


Fig. 1. Crystal structures of (a) ZnSiO_{3-m2}, (b) ZnSiO_{3-o}, and (c) ZnSiO_{3-r}.

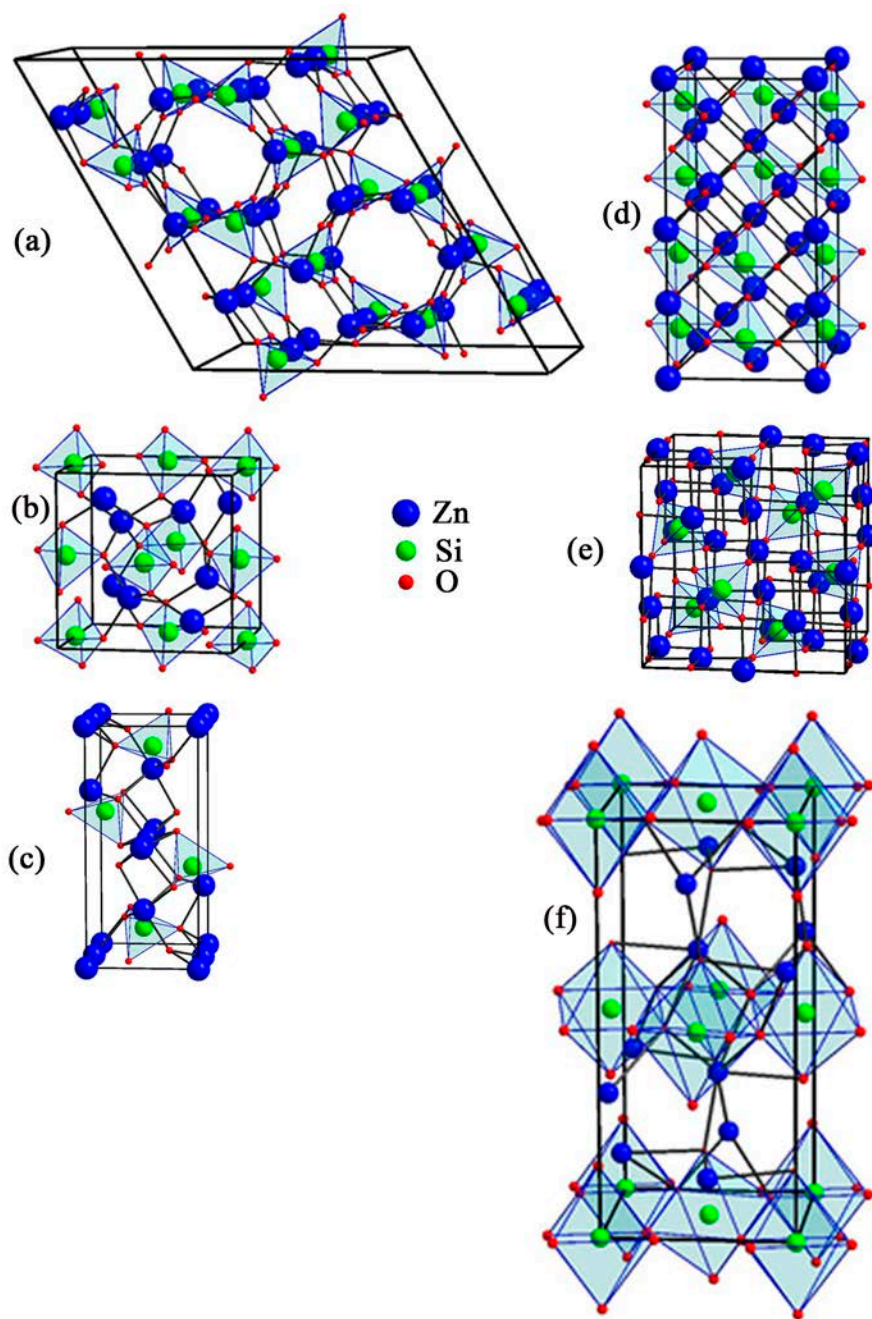


Fig. 2. Crystal structures of (a) Zn_2SiO_4-r , (b) Zn_2SiO_4-t , (c) $Zn_2SiO_4-o_1$, (d) $Zn_2SiO_4-o_2$, (e) Zn_2SiO_4-c , (f) $Zn_2SiO_4-o_3$.

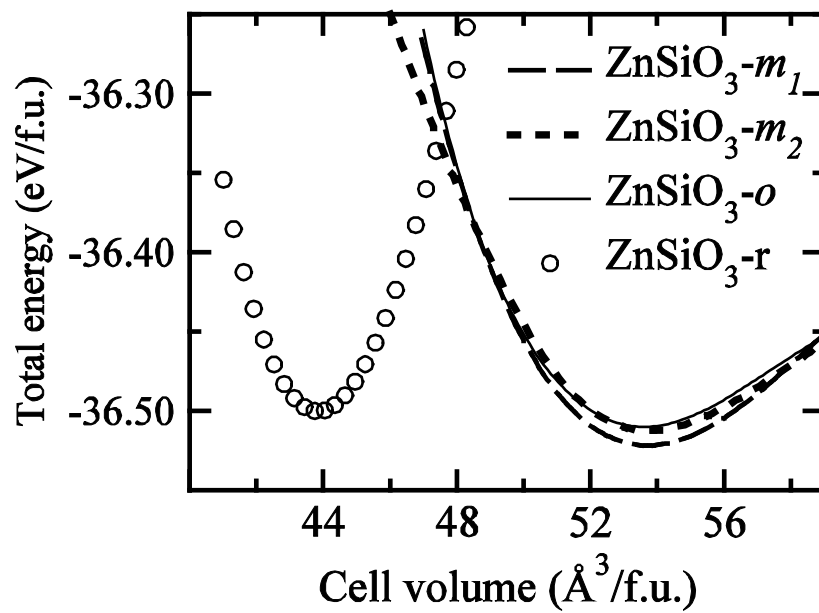


Fig. 3. Total energy vs volume curves for $\text{ZnSiO}_3\text{-}m_1$, $\text{-}m_2$, $\text{-}o$, and $\text{-}r$.

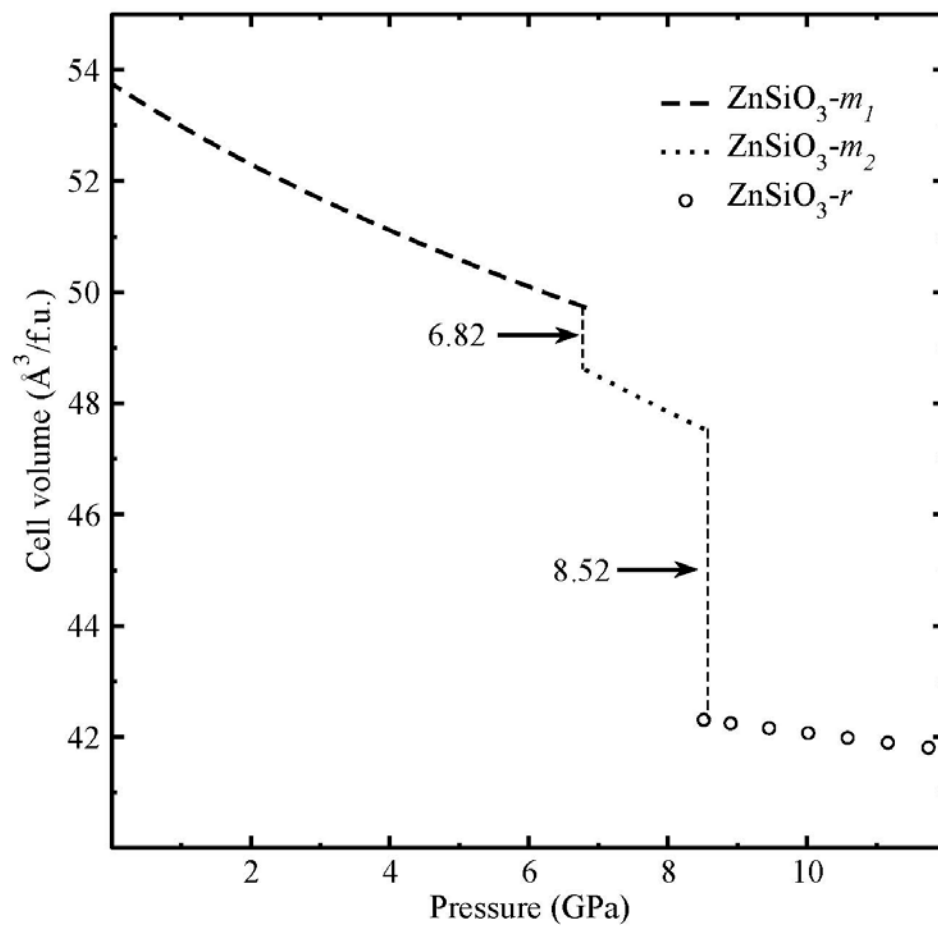


Fig. 4. Calculated cell volume vs pressure for ZnSiO_3 - m_1 , $-m_2$, and $-r$.

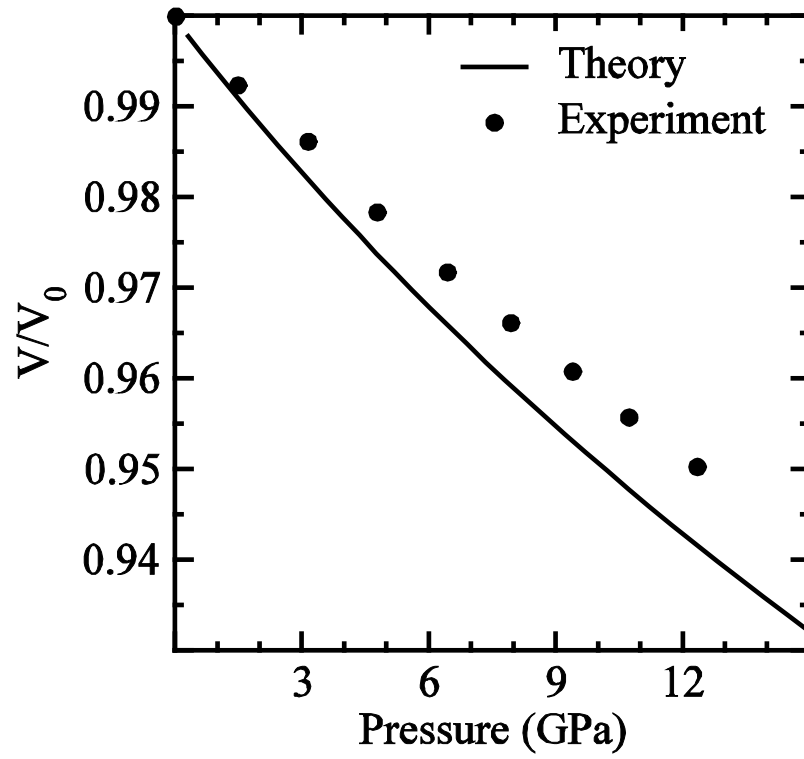


Fig. 5. Pressure dependence of the cell volume for ZnSiO_{3-r} along with the available experimental data from Ref. [(25)]

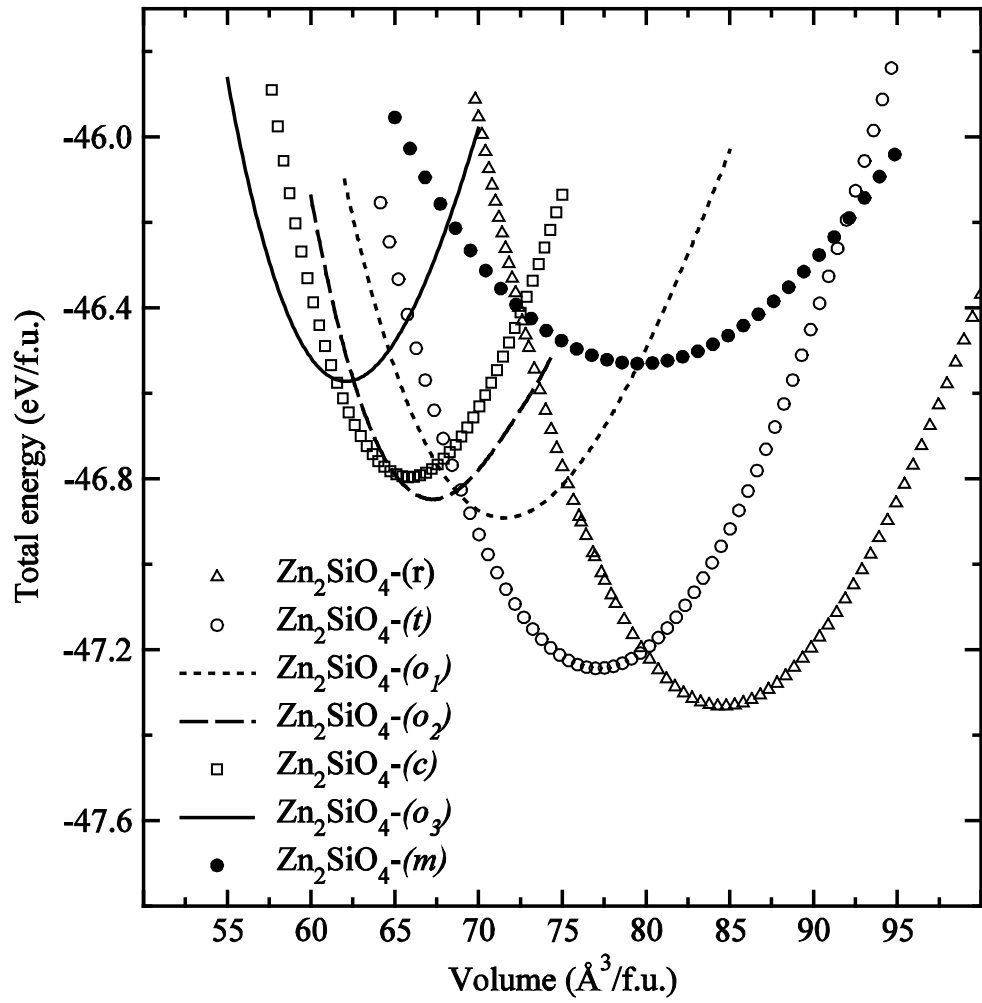


Fig. 6. Total energy vs volume for $\text{Zn}_2\text{SiO}_4\text{-}r$, $-t$, $-o_1$, $-o_2$, $-c$, $-o_3$, and $-m$.

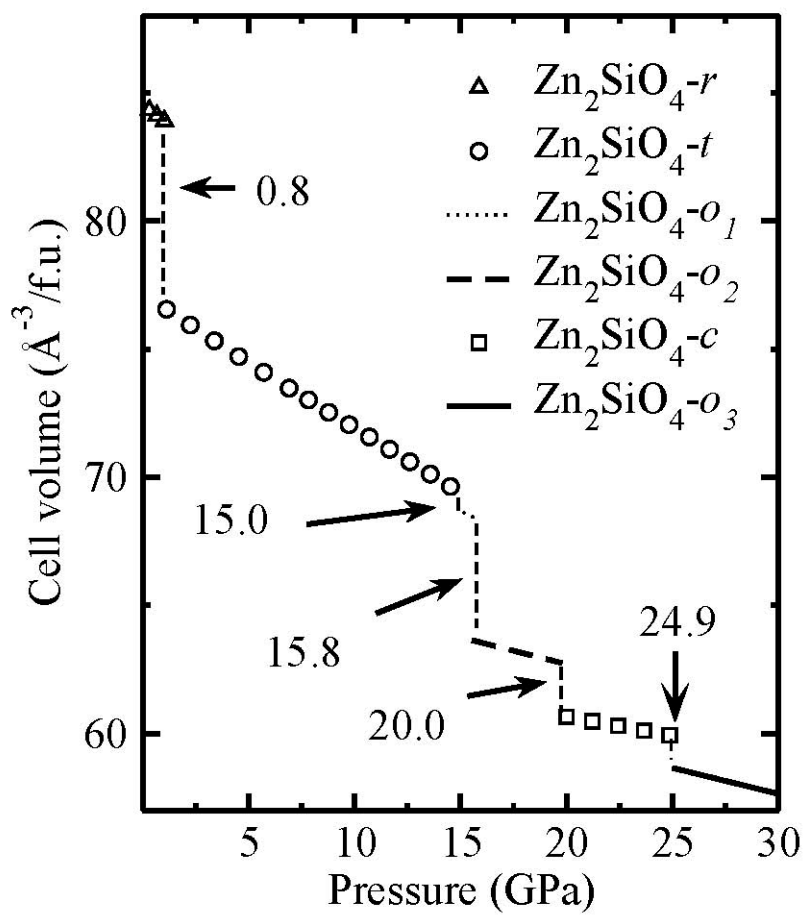


Fig. 7. Calculated cell volume vs pressure for $\text{Zn}_2\text{SiO}_4\text{-}r$, $\text{-}t$, $\text{-}o_1$, $\text{-}o_2$, $\text{-}c$, and $\text{-}o_3$.